

# Investigation of the Morphology of Surface Nanostructures by Means of the Synchrotron Radiation Based High-Resolution GEXRF Technique

S. Nowak<sup>1</sup>, Y. Kayser<sup>1</sup>, L. Baczewski<sup>3</sup>, D. Banaś<sup>2</sup>, W. Cao<sup>1</sup>, K. Deja<sup>2</sup>, J.-Cl. Dousse<sup>1</sup>, J. Hoszowska<sup>1</sup>, M. Pajek<sup>2</sup>, A. Petrouchik<sup>3</sup>, J. Szlachetko<sup>2,4</sup> and A. Wawro<sup>3</sup>

<sup>1</sup>Department of Physics, University of Fribourg, CH-1700 Fribourg, Switzerland

<sup>2</sup>Institute of Physics, Jan Kochanowski University, 25-406 Kielce, Poland

<sup>3</sup>Institute of Physics, Polish Academy of Sciences, 02-668 Warszawa, Poland

<sup>4</sup>European Synchrotron Radiation Facility, BP 220, F-38400 Grenoble, France



## Motivation

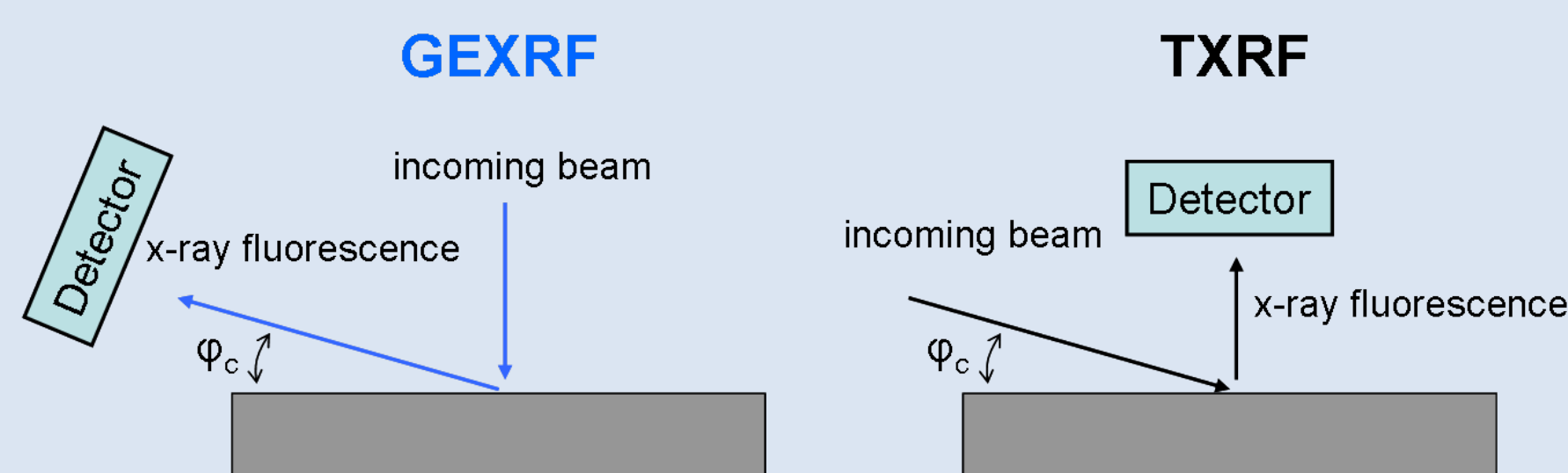
- Search for a multipurpose method of investigation of the surface structures
  - composition
  - 3D distribution
  - chemical state
  - morphology
- Test the utility of the GEXRF method applied to the investigation of the morphology of surface nanostructures

## Advantages of High-Resolution GEXRF

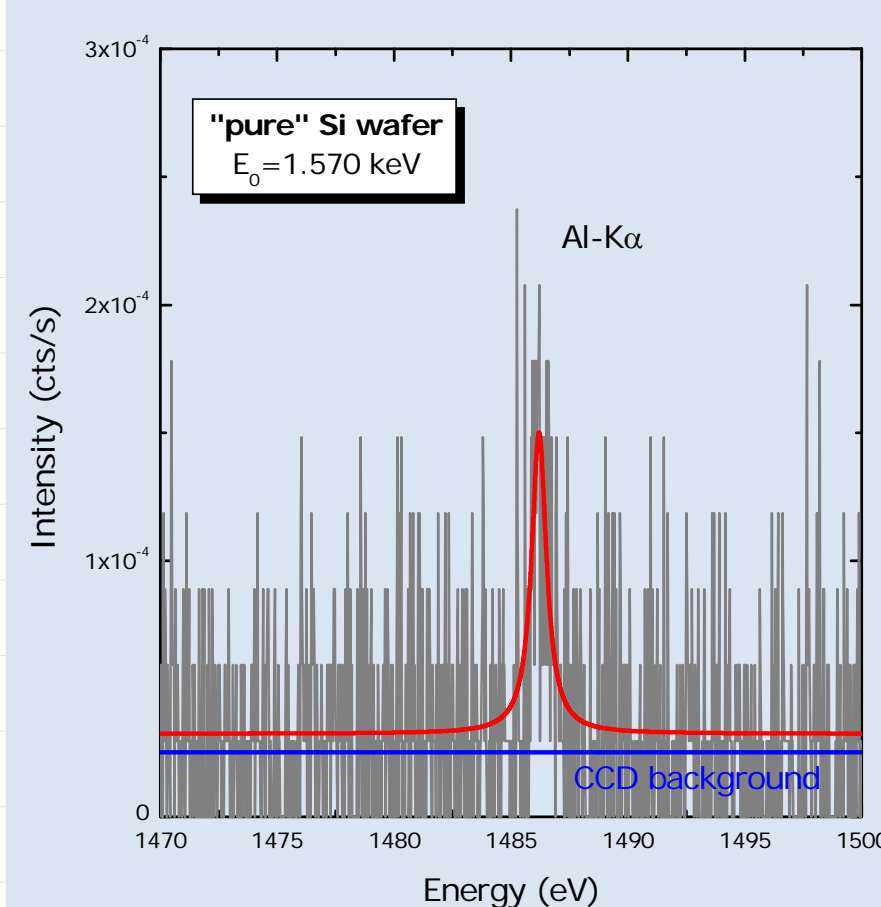
- Sample characterization (bulk, layer, islands)
- 2D-mapping with  $\mu\text{m}$  spatial resolution
- No overlapping of the fluorescence peaks — possibility of a better chemical specification
- Very good peak to background ratios (increased detection limits)

## Grazing Emission X-Ray Fluorescence (GEXRF)

- GEXRF — "Inversed" Total Reflection X-Ray Fluorescence (TXRF) [2]



- Detection of low-level surface impurities [3, 4]

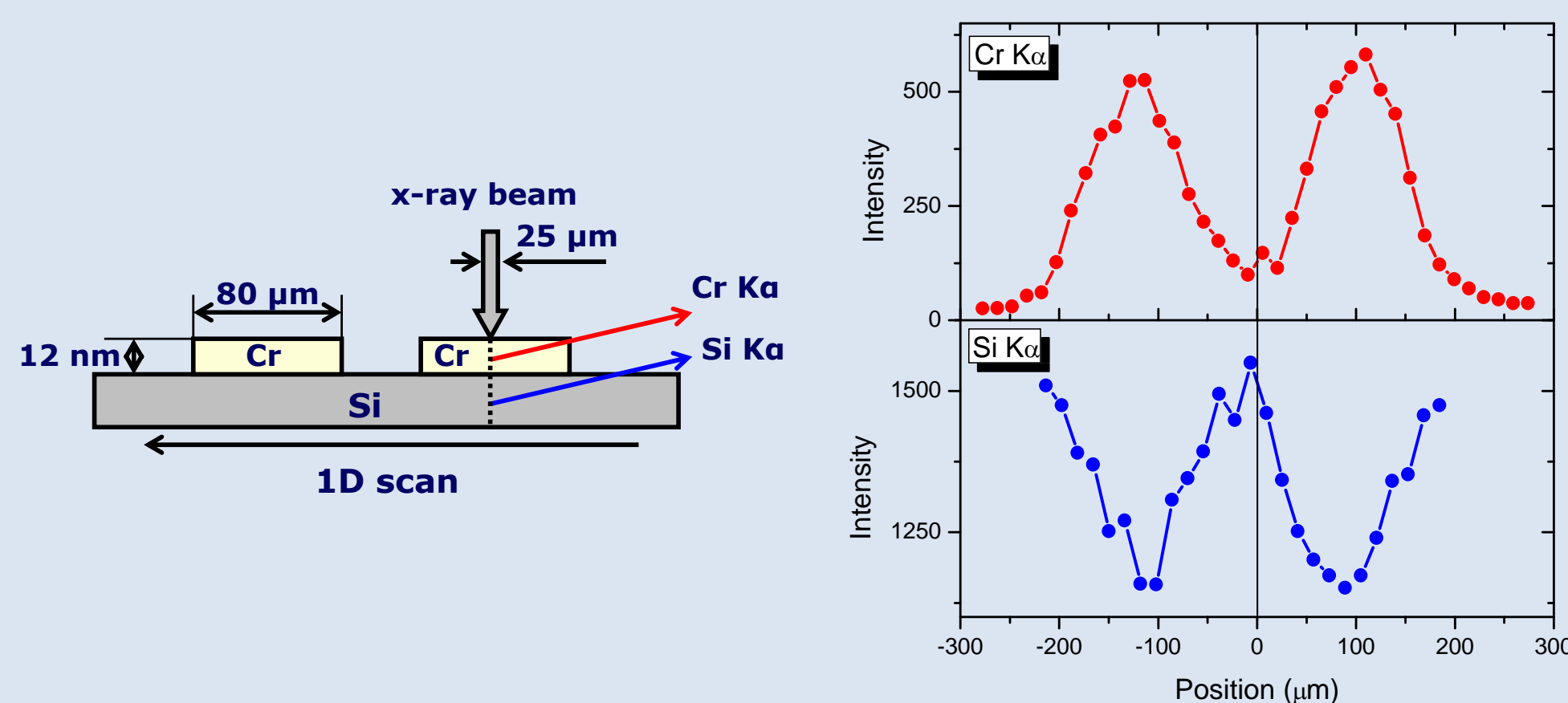


$$C_{dl}(1000s) = \frac{3\sqrt{N_{bcg}} \sqrt{t}}{N_{peak} \sqrt{1000s}} C$$

- Detection limit measured for Al impurities in Si substrate:

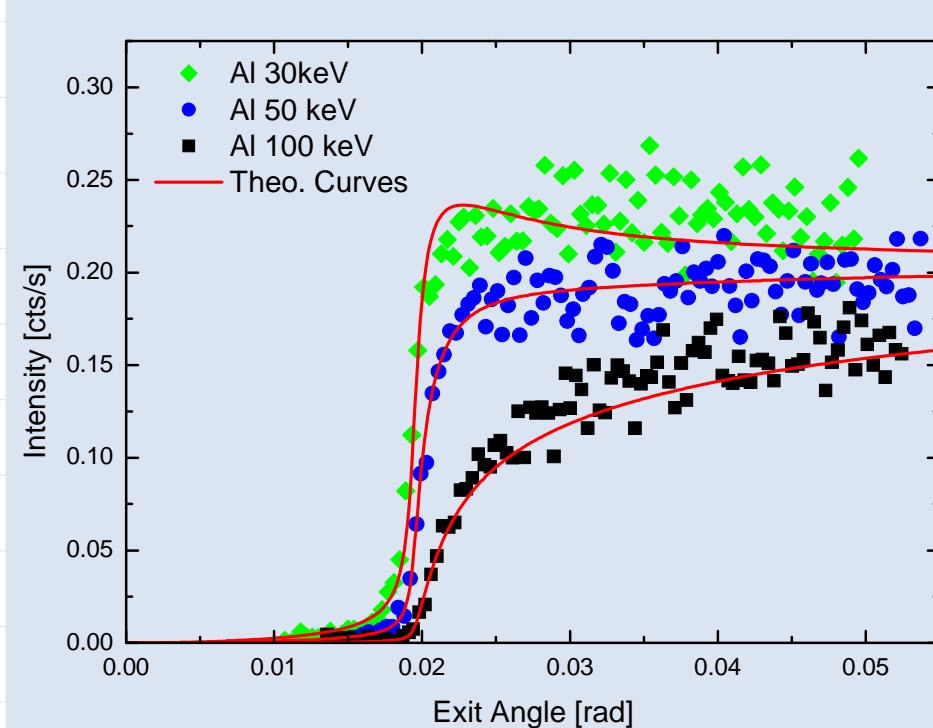
$$C_{dl} \approx 10^{12} \frac{\text{atom}}{\text{cm}^2}$$

- Lateral 2D-mapping [3]



Results of 1D scans across a pattern of Cr-strips deposited on a Si sample [3]

- Depth profiling [4]



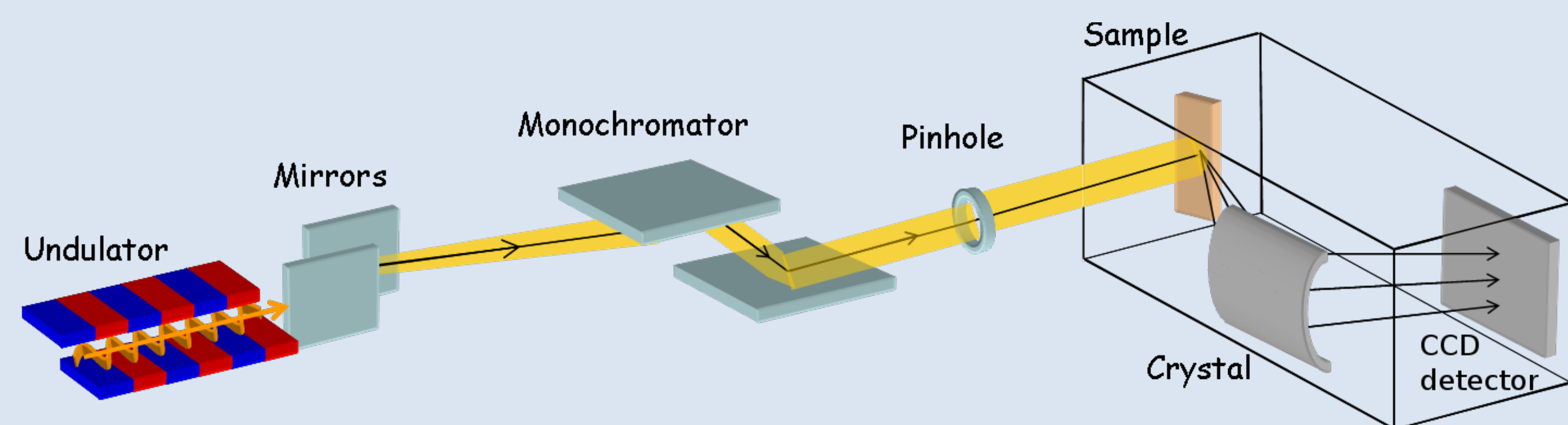
- Al ions implanted in the Si substrate with different ion energies
- Gaussian distribution of the implanted ions for the considered implantation energies
- Implantation dose of the Al ions:  $10^{16} \frac{\text{atom}}{\text{cm}^2}$

	Theo.	Exp.	Theo.	Exp.	Theo.	Exp.
Ion En. [keV]	30	35.47	50	48.73	100	97.21
Center [nm]	50.25	59.70	83.67	81.95	172.8	167.2
Width [nm]	23.38	26.30	35.56	34.64	61.93	60.77

## Experimental setup

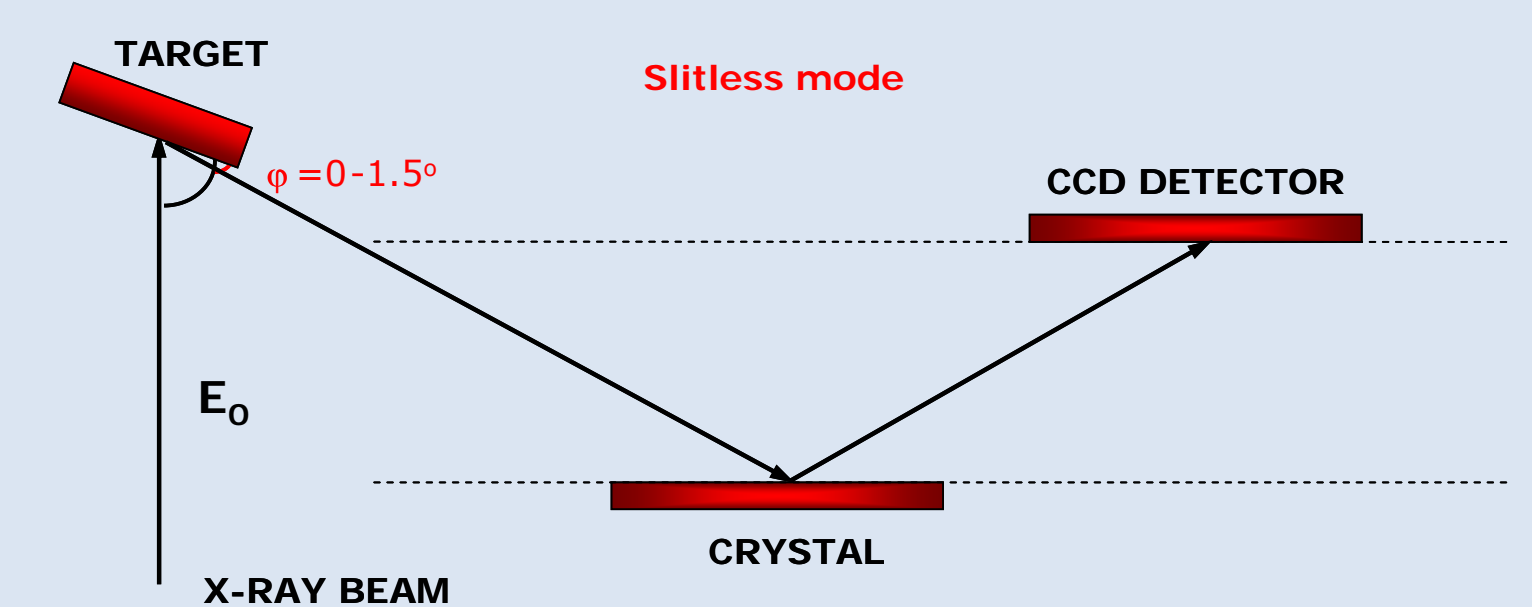
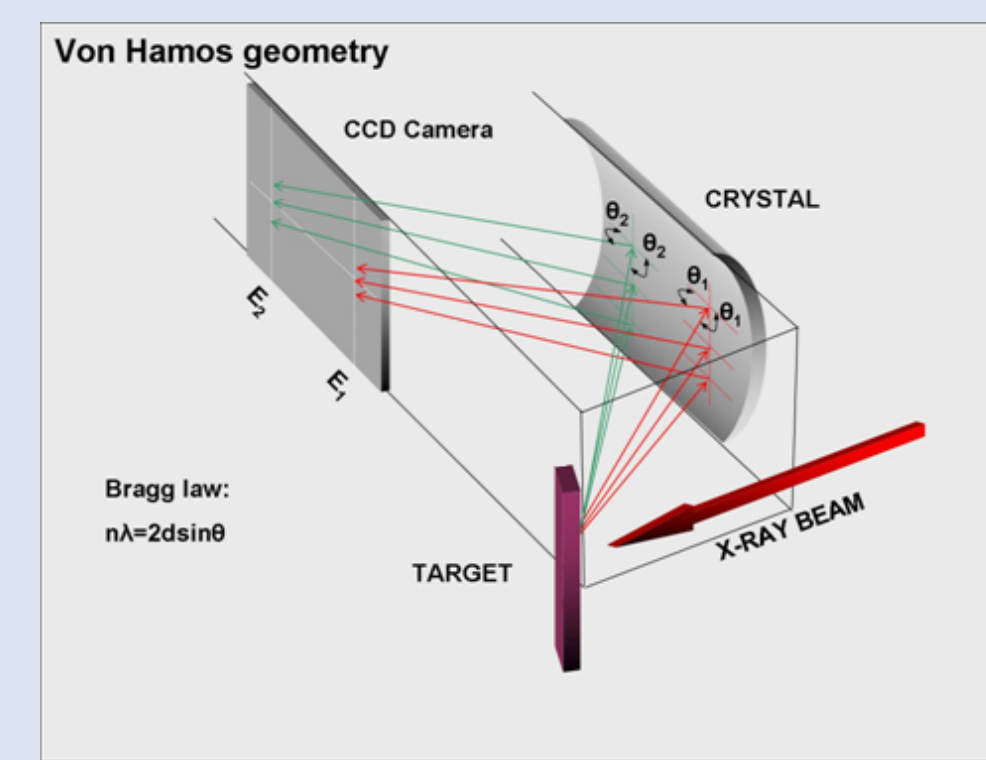
### Photon beam:

- $\sim 2 \text{ mm}^2$  beam size
- $\sim 10^{13}$  photons per second
- Energy range: 3.5 keV – 7.2 keV
- Few eV resolution (Ni/B<sub>4</sub>C multilayer monochromator)
- Horizontal polarization



### Von Hamos spectrometer [1]

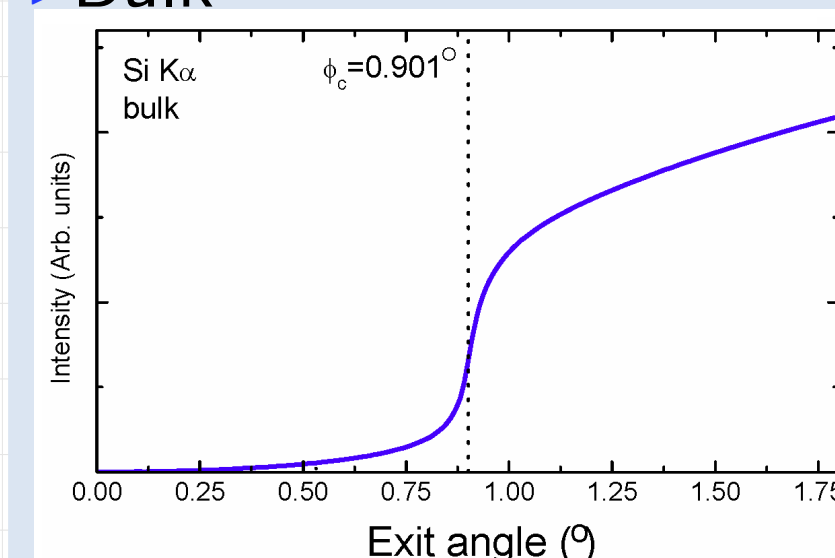
- Energy resolution:  $\sim 1 \text{ eV}$
- Back illuminated CCD camera
- Slitless operation mode
- Crystals: Ge(220), ADP(101)



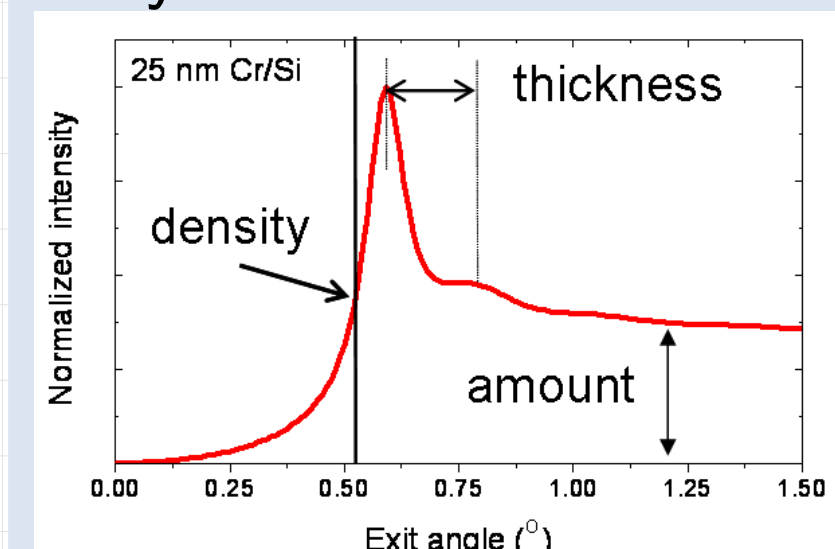
The high-resolution measurements were performed at the beamline ID21 of the European Synchrotron Radiation Facility (ESRF) in Grenoble, France.

## Different surface morphology measured by means of GEXRF [6]

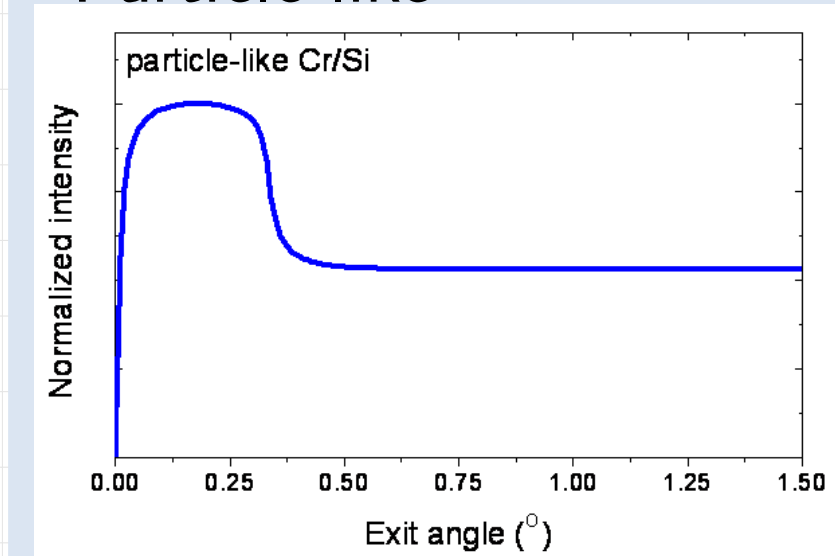
### Bulk



### Layer

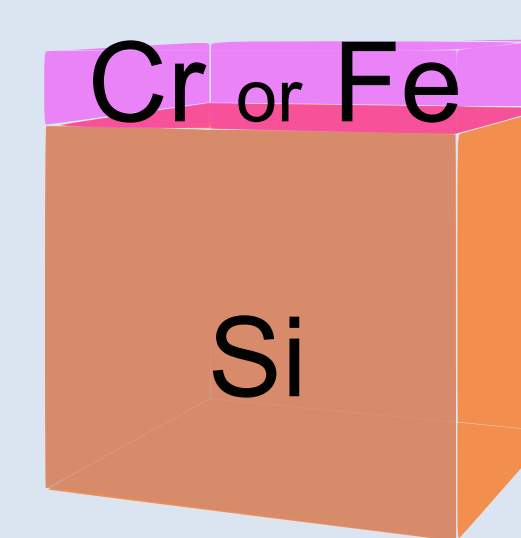


### Particle-like

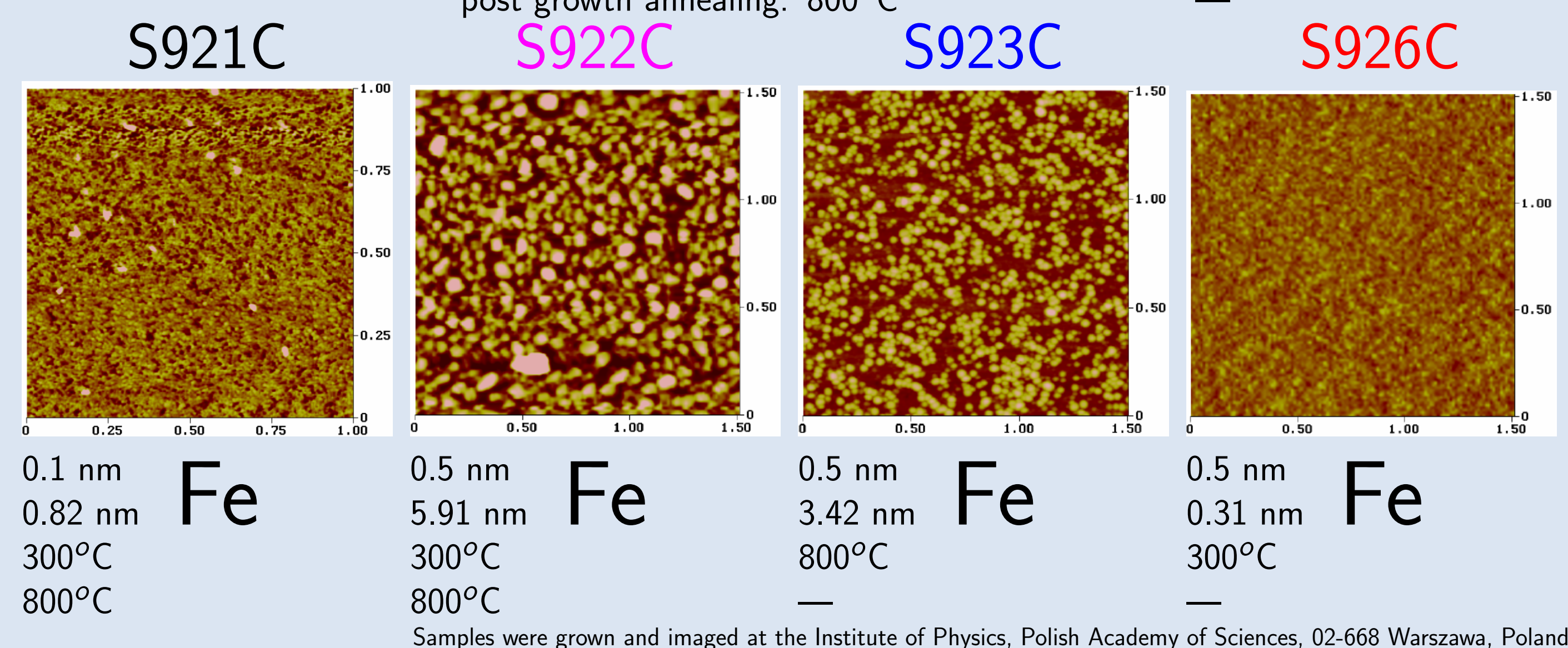
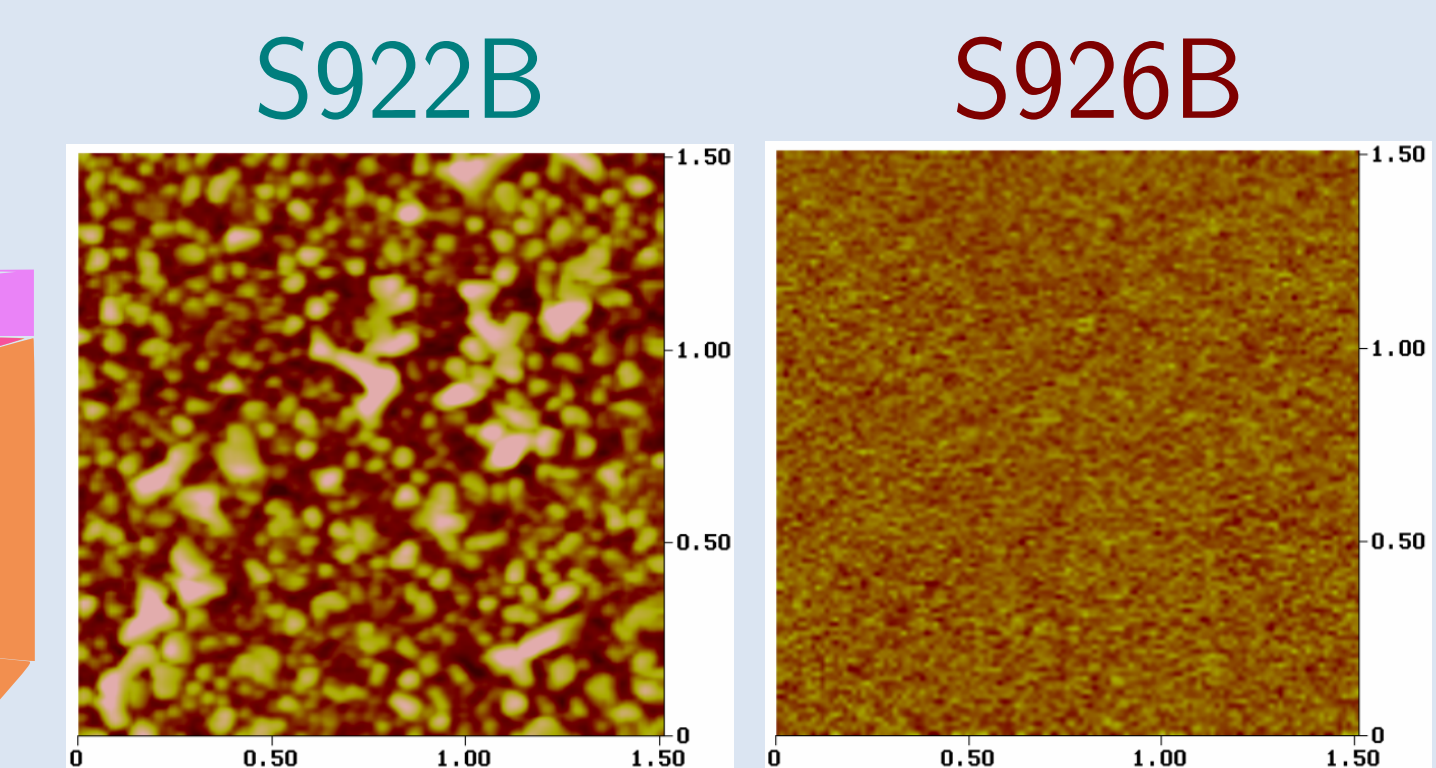


## Samples (AFM images)

- Si substrate
- Cr or Fe cap layer
- The same cap layer nominal thickness (for samples S922B, S926B and S922C, S923C, S926C)
- Big differences in the surface morphology — different growth conditions



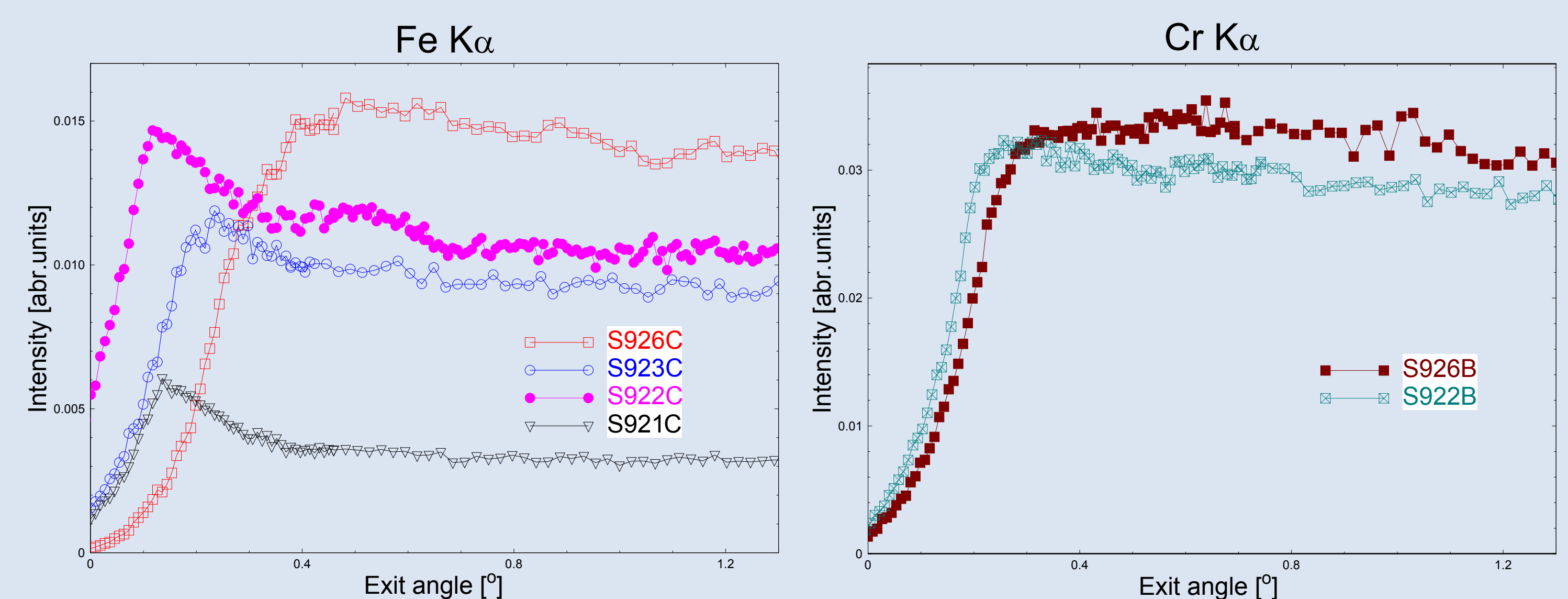
Cap thickness: 0.5 nm  
surface roughness: 4.07 nm  
growth temp.: 300°C  
post growth annealing: 800°C



Samples were grown and imaged at the Institute of Physics, Polish Academy of Sciences, 02-668 Warszawa, Poland.

## Results

- High sensitivity to changes of the surface morphology
- Layer-like angular dependences for large islands (sample S922B)
- Particle-like angular dependences for samples S921C and S922C
- Transition from particle-like to layer-like angular dependences (samples S922C, S923C and S926C)



## Conclusions

- Promising qualitative results
- Algorithm and its implementation for quantitative interpretation (height, width, separation of islands) is in development

The financial support of the Swiss National Science Foundation is acknowledged. The authors LTB, AW and AP are grateful for support from research funds of the Ministry of Science and Higher Education in Poland in the frame of the project N N507 452134 and the Polish National Scientific Network ARTMAG "Magnetic nanostructures for spintronics". The authors would also like to thank the ESRF for support.

## References

- J. Hoszowska et al., Nucl. Instrum. Meth. Phys. Res. 376, 129 (1996).
- Y. Kayser et al., EXRS Conference, 16-20 June 2008, Croatia, Book of abstracts p. 268.
- R. S. Becker, J. A. Golovchenko, and J. R. Patel, Phys. Rev. Lett. 50, 153 (1983).
- P. K. de Bokx et al. Spectrochim. Acta B 52, 829 (1997).
- J. Szlachetko et al., J. Appl. Phys. 105, 086101 (2009).
- H. P. Urbach and P. K. de Bokx, Phys. Rev. B 53, 3752 (1996).